isomers: \( \delta = 6.95 - 7.04 \) (m, PPh₃), 7.15 – 7.28 (m, NC₉H₆N), 7.82 – 7.90 (m, PPh₃); \(^1^3^B\) NMR spectra were not obtained due to low solubility; IR (Nujol): \( v = 3077.1(\text{m}), 1714(\text{s}), 1470(\text{m}) \text{ cm}^{-1} \); crystal dimensions 0.28 x 0.25 x 0.88 mm; monoclinic; space group \( \text{Cc} \); \( a = 30.0757(4), b = 0.6367(2), c = 17.1417(2), \beta = 111.176(1), Z = 8, V = 9902.82(5) \text{ Å}^3, \rho_{\text{calcd}} = 1.470 \text{ gm}^{-3}, T = 203 \text{ K}; 2\delta_{\text{max}} = 55^\circ \); of 3013 total reflections, 10794 were unique (\( R_{\text{e}} = 0.0259 \)); Siemens SMART CCD diffractometer, \( \text{MoK}_{\alpha} \) radiation (\( \lambda = 0.71073 \text{ Å} \); empirical absorption correction (\( \mu = 3.346 \text{ mm}^{-1}, T_{\text{min}}/T_{\text{max}} = 0.45/0.875 \)).

The structure was solved by Patterson and Fourier methods and refined by full-matrix least-squares on \( F^2 \) on all data. The asymmetric unit cell contains one molecule of \( 3 \) and one molecule of toluene; \( R_1 = 0.0247 \) for the 9473 observed data (\( I > 2 \sigma(I) \)) and \( \text{wR} = 0.0630 \) for all data. Goodness of fit on \( F^2 \) 1.035. Programs used: SHELXS-97 (structure solution) and SHELXL-97 (structure refinement). Crystallographic data (excluding structure factors) for the structures reported in this paper have been deposited with the Cambridge Crystallographic Data Centre as supplementary publication nos. CCDC-135052 (2b) and CCDC-135053 (3). Copies of the data can be obtained free of charge on application to CCDC, 12 Union Road, Cambridge CB2 1EZ, UK (fax: (+44) 1223 336 033; e-mail: deposit@ccdc.cam.ac.uk).

[8] Longest of the 629 determinations found in the CSD.

[9] \( t \) denotes that the signal has apparent multiplicity; \( vJ(\text{CP}) \) is the sum of the two coupling constants \( J(\text{CP}) + J(\text{CP}) \), as explained in S. M. Maddock, C. E. F. Rickard, W. R. Roper, L. J. Wright, Organometallics 1996, 15, 1793 – 1803.

**Orientation-Controlled Monolayer Assembly of Zeolite Crystals on Glass and Mica by Covalent Linkage of Surface-Bound Epoxy and Amine Groups**

Alexander Kulak, Yun-Jo Lee, Yong Soo Park, and Kyung Byung Yoon*

Efforts have been made during the last decade to develop zeolite thin films for separative, catalytic, and chemical sensor applications.\(^1\)\(^-\)\(^10\) For practical purposes, they have usually been prepared on various substrates or supports since they are extremely fragile. The chemical methods to assemble zeolite thin films on substrates can generally be classified into two groups, namely, “direct growth”\(^1\)\(^-\)\(^15\) and “postsynthetic crystal attachment”.\(^6\)\(^-\)\(^7\) The former group usually involves in situ growth of the zeolite crystals or films on the substrates by immersing them into the synthesis gel.\(^1\)\(^-\)\(^10\) For best results, the substrates are often chemically modified or pretreated with colloidal seed crystals. The unique pulsed laser ablation of preformed supported zeolites can also be classified into this group.\(^5\) In the latter group, preformed zeolite crystals are chemically attached on substrates by ionic or covalent bonding by use of appropriate organic additives.\(^6\)\(^-\)\(^7\) For these
methods, the organic functional groups are usually tethered onto either the zeolites or the substrates by covalent bonding, proceeded by subsequent chemical linkage between the tethered functional groups and the unmodified surface. However, the coverage and orientation of the zeolite crystals attached onto the substrates by this “one-component modification” were rather poor. In fact, the ability to uniformly align zeolite crystals on substrates during the assembly is essential for the applications of zeolite thin films as the hosts for quantum dots, magnetic particles, nonlinear optical materials,[8, 9] or chemical sensors.[11] It has been reported that zeolite crystals tend to physically adhere onto substrates with their flat faces facing the substrate when the substrate is dip coated into a colloidal zeolite solution.[2] This phenomenon led to uniform alignment of cubic zeolite A crystals on substrates with a face parallel to the plane of the substrates.[2] For zeolites with lower symmetry, particularly for those that possess significant net intrinsic dipole moments due to one of the axes being longer than the other two, the application of strong electric fields is highly effective in aligning the crystals with the longest axis parallel to the direction of the electric field.[10] We now report a novel “two-component modification” strategy that leads to monolayer assembly of zeolite A and ZSM-5 crystals onto glass and mica with very high surface coverage and also uniform alignment. This new approach involves independent tethering of two different functional groups onto the zeolite and the substrate, followed by covalent linking of the two tethered functional groups as schematically illustrated in Scheme 1.

Visual observation of the zeolite-coated glass plate prepared by the method shown in Scheme 1 revealed that the entire glass plate (18×18 mm) was uniformly covered with white zeolite A crystals. A thin layer of zeolite particles maintained adhesion onto the glass substrate even after sonication for 30 s in toluene. Omitting the surface anchoring of any of the silyl compounds in the binding procedure resulted in essentially empty glass plates scarcely covered with physisorbed zeolite crystals; these few crystals immediately fell off the glass plate during sonication. This result strongly indicates that facile formation of amine–hydroxide bridges indeed proceeds between terminal amine and epoxide groups, which leads to effective assembly of zeolite A crystals on the glass substrate. The efficacy of uniform coverage of the entire glass surface seemed to depend on factors such as the uniformity of the chemical composition, the pretreatment of the glass substrate, the degree of coverage of the corresponding organic compound on the glass and zeolite, and the size and morphology of the zeolite crystals.

The scanning electron micrograph (SEM) of the glass plate coated with zeolite A crystals revealed that almost the entire plate was covered by the zeolite crystals with high-density packing, as shown in Figure 1A. Observation at a higher magnification of Figure 1B revealed that the entire glass plate was covered with a dense layer of zeolite A crystals. The scanning electron micrograph (SEM) of the glass plate coated with ZSM-5 crystals revealed that the entire plate was covered by a dense layer of ZSM-5 crystals, as shown in Figure 1C.

Scheme 1. The procedure to assemble zeolite A monolayer on a glass substrate.
magnification (Figure 1B) revealed that only a monolayer of the crystals was assembled on the glass. Moreover, this image shows the striking feature that all the zeolite crystals are aligned with a face parallel to the glass substrate, a perfect one-dimensional orientation.

Most interestingly, the zeolite A crystals showed a strong tendency to pack closely during the attachment onto the substrate with the side faces in contact. This phenomenon indicates that a 3-aminopropyl-covered zeolite A crystal, once mounted on the substrate, tends to act as a template for positioning the next crystal. Although further study is necessary, we propose for the moment that such a close packing arises due to a strong face-to-face interaction between the 3-aminopropyl-covered faces of two separate zeolite A crystals through a large number of hydrogen bonds between the terminal NH₂ groups. Consistent with our proposal, the commonly observed close packing of the physisorbed, unmodified zeolite crystals on substrates during dip coatings has also been attributed to hydrogen bonding between the surface hydroxyl groups of zeolites.[2]

Figure 1 C shows the SEM image of a monolayer of ZSM-5 crystals similarly assembled on a glass substrate. This figure demonstrates that the method described above works equally well even for large (≈5 μm) zeolite crystals. This image further shows that all the ZSM-5 crystals assemble with their b axis perpendicular to the glass surface despite that the rectangular-shaped side faces can, in principle, also be attached onto the surface. This preferential orientation might occur because ZSM-5 crystals can have better contact with the glass substrate through their substantially larger [010] face than through their smaller side faces. It was also noticed that the intercrystalline void spaces increase upon changing from zeolite A to ZSM-5, presumably due to the lower symmetry and increased size of ZSM-5 relative to zeolite A. The orthorhombic CoAPO-34 and SAPO-34 molecular sieve crystals (≈7 μm) also efficiently assembled onto the glass substrate with similar high-density, monolayer packing by the same method. The SEM images of the assembled molecular sieves (not shown) revealed that these crystals also align with a face parallel to the substrate. However, no preference for a particular face for these crystals was observed, presumably due to less substantial differences in the area of the three faces. The attachment method was also effective for freshly and increased size of ZSM-5 relative to zeolite A. The orthorhombic CoAPO-34 and SAPO-34 molecular sieve crystals (≈7 μm) also efficiently assembled onto the glass substrate with similar high-density, monolayer packing by the same method. The SEM images of the assembled molecular sieves (not shown) revealed that these crystals also align with a face parallel to the substrate. However, no preference for a particular face for these crystals was observed, presumably due to less substantial differences in the area of the three faces. The attachment method was also effective for freshly cut mica as the substrate, although the coverage was usually a bit less than onto glass.

The XRD pattern of the zeolite A-coated glass showed only five lines that correspond to [200], [600], [800], [1000], and [1200] planes of the zeolite A crystals (Figure 2a). The reflection for [400] expected at 2θ = 14.4° seems to be lost in the background due to an intrinsically very low intensity (less than 0.5 % of the [200] reflection).[12] Such a simple pattern contrasts strongly with the more common, complex pattern of the randomly oriented zeolite A crystal powder shown in the inset. Likewise, only five lines appeared from the monolayer of ZSM-5 crystals due to [020], [040], [060], [080], and [0100] reflections (Figure 2b).

The strength of bonding between zeolite A crystals and substrates seems to vary and is sensitively dependent on the assembly procedure. That a large number of zeolite A crystals remain adhered to the glass substrate even after sonication as long as 5 min suggests the bonding to be fairly strong. Interestingly, SEM images of the extensively sonicated samples revealed that most of the surviving zeolite particles had become severely fractured. This result contrasts with the fact that all the zeolite A particles retain their original cubic morphology under the same treatment conditions as a free powder. The larger ZSM-5, CoAPO-34, and SAPO-34 crystals tend to be more easily removed by sonication.

This report introduces a novel strategy to assemble monolayers of zeolite A and ZSM-5 onto glass and mica substrates with a perfect one-dimensional orientation through covalent binding between two independently tethered, terminal functional groups, in this case epoxide and amine. This method uniquely provides strong zeolite–substrate bonding, high surface coverage, uniform orientation, and easy control of the monolayer assembly. In a broader sense, this report demonstrates a novel strategy to efficiently organize micrometer-sized nanoporous inorganic crystals into functional macroscopic structure.

**Experimental Section**

Zeolite A, ZSM-5, CoAPO-34, and SAPO-34 were synthesized according to the literature procedures. The synthetic templates were not removed from the zeolite or molecular sieve crystals prior to assembly. The washed and dried zeolite or molecular sieve (0.2 g) was treated with 3-aminopropyltriethoxysilane (2 mM, 10 mL, 110 °C, 1 h) under argon. This led to 3-aminopropylsilyle groups covalently bound onto the zeolite or molecular sieve surface (see Scheme 1A). The presence of the surface-bound 3-aminopropylsilyle groups was confirmed by purple coloration of the surface when treated with 1,2,3-triketohydridenedimethyldihydroxamine. Control experiments with
fresh zeolites undoped with aminopropylsilyle groups did not develop the characteristic purple color.

Independently, a piece of thin glass (18 × 18 mm²) was treated with 3-(2,3-epoxypropoxy)propyl]trimethoxysilane (2 mmol) in toluene (10 mL, 110 °C, 1 h) under argon to assemble a layer of 3-(2,3-epoxypropoxy)propylsilyle groups on the glass surface (see Scheme 1B). The presence of surface-bound terminal epoxy groups attached on the glass was confirmed by pale-colored solution of the reaction mixture and washed extensively with toluene for 20 s to remove physisorbed zeolite crystals on the chemically bound first layer.

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Addition of Enantioenriched γ-Oxygenated Allylic Stannanes to N-Acyl Iminium Intermediates: A New Synthesis of syn-Amino Alcohol Derivatives**

James A. Marshall,* Kevin Gill, and Boris M. Seletsky

Some years ago we discovered a facile route to monoprotected syn-1,2-diol derivatives through BF₄⁻-promoted addition of enantioenriched γ-oxygenated allylic stannanes to aldehydes (Scheme 1).[1] We had hoped to extend these

![Scheme 1. A facile route to monoprotected syn-1,2-diol derivatives](image)

additions to imines, along the lines reported by Keck and Enholm (Scheme 2),[2] but were unsuccessful in these attempts. No detectable β-amino ether adducts were formed, even at room temperature. We attributed these failures to the lower reactivity of oxygenated allylic stannanes relative to their nonoxygenated allyl and crotyl counterparts.[3]

![Scheme 2. Addition of allylic stannanes to imines](image)

A report by Yamamoto and Schmid, describing the addition of a γ-OMOM allylic stannane to several N-acyliminium intermediates from Hiemstra and Speckamp,[4] prompted our examination of this alternative route to β-amino alcohol derivatives.[5] In fact, addition of the racemic (Z)-γ-oxygenated allylic stannanes 2a and 2b[6] to the N-acyliminium precursor 1, derived from isovaleraldehyde,[3] proceeded in high yield to afford the desired adducts (Table 1). Unfortunately, a mixture of syn and anti isomers 3 and 4 was obtained from these additions.

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[1] Prof. J. A. Marshall, K. Gill
Department of Chemistry
University of Virginia
Charlottesville, VA 22904 (USA)
Fax: (+1) 804-924-7993
E-mail: jama5x@virginia.edu
Dr. B. M. Seletsky
Eisai Research Institute
4 Corporate Drive, Andover, MA 01910 (USA)

Supporting information for this article is available on the WWW under http://www.wiley-vch.de/home/angewandte/ or from the author.

* To whom correspondence should be addressed.